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## Nonaqueous Synthesis and Selective Crystallization of Gallium Sulfide Clusters into Three-Dimensional Photoluminescent Superlattices

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Despite extensive research on chalcogenide clusters and open frameworks,<sup>1–8</sup> little is known about related gallium sulfides.<sup>9</sup> Most known gallium sulfides are made by high-temperature melt or solid-state reactions. As a result, few three-dimensional (3D) open-framework gallium sulfides are known.

We have now developed a low-temperature and nonaqueous synthesis approach and have successfully prepared a series of gallium sulfide and polysulfide open-framework materials built from binary Ga-S and ternary Zn-Ga-S clusters (Table 1).<sup>10</sup> These materials possess unprecedented structural properties, and their fluorescent emission wavelength fills the gap between open-framework oxides and indium chalcogenides.

All new compounds contain supertetrahedral clusters that are tetrahedrally shaped fragments of the cubic ZnS-type lattice. They are denoted as Tn by Yaghi, O'Keeffe et al.<sup>11,12</sup> With Tn clusters as artificial tetrahedral atoms (T-atoms), it is possible to construct covalent superlattices with framework topology similar to that in zeolites. However, the ring size in terms of the number of T-atoms is increased by n times. An increase in the ring size is important because crystalline porous materials with a ring size larger than 12 T-atoms are scarce but highly desirable for applications involving large molecules.

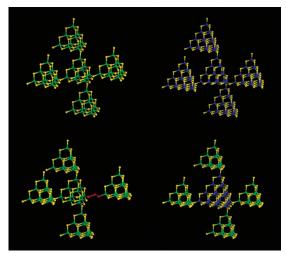
In the absence of divalent cations, one of the superlattices formed is UCR-7GaS-TETA (Figure 1) built from T3 (Ga<sub>10</sub>S<sub>18</sub><sup>6-</sup>) clusters. All four corners of T3 clusters are shared through S<sup>2-</sup> bridges. The T3 cluster is known in both divalent metal thiolates (e.g., Cd<sub>10</sub>S<sub>4</sub>(SPh)<sub>16</sub><sup>4-</sup>) and boron or indium sulfides.<sup>1,11</sup> However, prior to this work, neither the molecular T3 cluster nor its covalent superlattice is known in the Ga–S system.

When divalent cations are introduced either directly through the addition of a salt (e.g.,  $Zn(NO_3)_2 \cdot 6H_2O$ ) or through the oxidation of the elemental form (e.g., Zn), the reaction system gains an extra capability to form T4 clusters ( $Zn_4Ga_{16}S_{33}^{10-}$ ), in addition to its inherent ability to form T3 clusters ( $Ga_{10}S_{18}^{6-}$ ). This is because regular Tn (n > 3) clusters can only be found in heterometallic compositions due to the local charge balance requirement around the four-coordinated core sulfur atoms.<sup>12–14</sup>

Therefore, both T3 and T4 clusters are likely present in the heterometallic sulfide solution. By using different structure-directing agents, one or both of them can be crystallized into a superlattice. In this work, several situations could be distinguished, leading to the synthesis of a number of superlattices built from binary and ternary sulfide clusters.

The first situation involves the use of 1-(2-aminoethyl)piperazine (AEP, C<sub>6</sub>H<sub>15</sub>N<sub>3</sub>). Only T3 clusters are formed in crystals (denoted as UCR-18GaS-AEP), even if various amounts of Zn<sup>2+</sup> cations are present. An unprecedented feature is the presence of the polysulfide linkage between T3 clusters. This is the first observation of





**Figure 1.** Four structure types made from T3 and T4 clusters. (Top left): the T3–T3 superlattice in UCR-7; (top right): the T4–T4 superlattice in UCR-5; (bottom left): the T3–T3 superlattice through mixed -S- and -S-S-S- bridges in UCR-18; (bottom right): the hybrid T3–T4 superlattice in UCR-19.

supertetrahedral clusters being linked through polysulfide bonds. Here, T3 clusters are connected at three corners through  $S^{2-}$  bridges into a 3-connected sheet with 6-rings (six T3 clusters in a ring). These sheets are joined through -S-S-S- bridges into a 3D net.

In comparison, AEP gives a regular T3 superlattice in the In–S system with only S<sup>2–</sup> bridges (Table 1). In the Ga–S system, it appears that the lattice contraction caused by smaller Ga<sup>3+</sup> cations makes it less accommodating to AEP molecules. As a result, the introduction of longer S<sub>3</sub><sup>2–</sup> groups in UCR-18GaS-AEP further pushes the clusters apart to offset the contraction effect of Ga<sup>3+</sup> cations.

The selection of T4 clusters can be achieved with 1,4-bis(3aminopropyl)piperazine (BAPP,  $C_{10}H_{24}N_4$ ). In this case, UCR-5ZnGaS-BAPP (Figure 1) built from T4 clusters is formed. Compared to AEP with 9 non-H atoms, BAPP has 14 non-H atoms. The large size of BAPP might be one reason for the preferential selection of large T4 clusters that once crystallized give a larger pore volume.

An interesting situation occurs when triethylenetetramine (TETA,  $C_6H_{18}N_4$ ) (10 non-H atoms) is used. In the absence of  $Zn^{2+}$ , it forms a regular T3 structure (UCR-7GaS-TETA). However, when  $Zn^{2+}$  cations are introduced, UCR-19ZnGaS-TETA (Figure 2) with alternating T3 and T4 clusters is formed. UCR-19ZnGaS-TETA is the only known hybrid superlattice containing different regular supertetrahedral clusters. Prior to this work, the T3 cluster is known to connect with the coreless pseudo-T5 cluster into a superlattice.<sup>15</sup>

All materials reported here have a framework topology resembling that of the zinc blende when Tn clusters are represented as

Table 1. Summary of Important Data for Selected Structures Synthesized in This Study

name <sup>a</sup>	framework composition	space group	a (Â)	<i>c</i> (Â)	$R(F)^{b}$	$\lambda_{ m em}$ (nm)	$\lambda_{\mathrm{ex}}$ (nm)
UCR-18GaS-AEP <sup>b</sup>	$[Ga_{10}S_{17,5}(S_3)_{0,5}]^{6-}$	C2/c	35.903(5)	21.183(3)	6.41	500	375
UCR-19ZnGaS-TETA	$[Ga_{10}S_{18}Zn_4Ga_{16}S_{33}]^{16-}$	$I4_1/a$	21.087(2)	35.582(6)	6.08		
UCR-5ZnGaS-BAPP	$[Zn_4Ga_{16}S_{33}]^{10-}$	$I4_1/acd$	22.595(3)	40.905(8)	6.71	440	368
UCR-7GaS-TETA	$[Ga_{10}S_{18}]^{6-}$	$I4_1/acd$	19.201(2)	29.815(4)	5.85	476	370
UCR-7GaS-TAEA	$[Ga_{10}S_{18}]^{6-}$	$I4_1/acd$	19.220(3)	29.866(6)	5.80		
UCR-7GaS-DBA	$[Ga_{10}S_{18}]^{6-}$	$P4_{1}2_{1}2$	18.803(4)	29.591(8)	9.92		
UCR-7GaInS-TETA	$[Ga_{4.5}In_{5.5}S_{18}]^{6-}$	$I4_1/acd$	19.715(2)	30.996(4)	5.87	457	367
UCR-7InS-AEP	$[In_{10}S_{18}]^{6-}$	$I4_1/acd$	20.318(2)	31.994(5)	4.74		

<sup>a</sup> TAEA = Tris(2-aminoethyl)amine,  $C_6H_{18}N_3$ ; DBA = di-*n*-butylamine. In all compounds, framework atoms are located, and guest amine molecules are disordered.  ${}^{b}R(F) = \sum ||F_{o}| - |F_{c}||/\sum |F_{o}|$  with  $F_{o} > 4.0\sigma(F)$ . For UCR-18GaS-AEP, b = 18.501(3) Å,  $\beta = 115.992(3)^{\circ}$ .

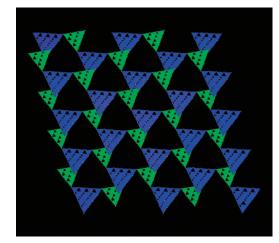


Figure 2. 3D framework of the T3-T4 hybrid framework in UCR-19ZnGaS-TETA. Green: T3 clusters. Blue: T4 clusters. Only one set of superlattice is shown for clarity.

artificial T-atoms. Even though there are two interpenetrating lattices, the large size of Tn clusters leads to a highly open framework. In all compounds, disordered extra-framework species occupy 56% of the crystal volume or more as calculated with PLATON.<sup>16</sup>

These gallium sulfide superlattices serve to bridge a previously observed gap in the emission wavelength of open-framework phosphors. It was reported earlier that open-framework oxides have an emission wavelength from 400 to 440 nm.17 Later, openframework indium sulfides were found to emit in the range from 520 to 570 nm.14,18 Gallium sulfides reported here show strong photoluminescent emissions that can be varied from 440 to 500 nm (Table 1). For example, UCR-18GaS-AEP can be excited by a broad spectral range from 300 to 470 nm and emits at 500 nm (fwhm  $\approx 100$  nm).

Gallium sulfide superlattices reported here are more thermally stable than corresponding indium sulfides. For UCR-7GaS-TETA, after heating at 280 °C in flowing argon, over 49% of H and 63% of N were removed.<sup>10</sup> The XRD shows no deterioration in the sample crystallinity. UCR-7GaS-TETA remained crystalline after being heated at 300 °C in air, while UCR-7InS-AEP was found amorphous under the same condition.

During the direct calcination of UCR-7GaS-TETA, the amount of carbon was not reduced, likely because of the coke formation.<sup>10</sup> To help remove carbon and also to examine ion-exchange properties, UCR-7GaS-TETA was subject to ion exchange with 2 M NH<sub>4</sub>-Cl solution. The elemental analysis showed that over half of carbon was removed after 6 d at room temperature.<sup>10</sup> The XRD showed that the ion-exchanged sample did not suffer a loss of crystallinity. Similar ion-exchange reactions were also observed for other monoor divalent cations such as Na<sup>+</sup> and Ca<sup>2+</sup>.

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Supporting Information Available: Crystallographic data including positional parameters, thermal parameters, and bond distances and angles (CIF). This material is available free of charge via the Internet at http://pubs.acs.org.

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- (10) A typical synthesis is given below for the preparation of UCR-18GaS-AEP. Gallium metal (0.1266 g), sulfur (0.1921 g), 1-(2-aninoethyl)-piperazine (2.0246 g) were mixed in a 23-mL Teflon-lined stainless steel autoclave. The vessel was sealed and heated at 190 °C for 6 d. The autoclave was allowed to cool to room temperature. Light-yellow prismatic crystals were obtained in 86% yield. Elemental analysis (wt %) for UCR-18GaS-AEP: 15.25 C, 3.70 H, 9.35 N, calculated: 15.75 C, 3.64 H, 9.19 N based on the formula  $[Ga_{10}S_{17.5}(S_3)_{0.5}](C_6H_{15}N_3H)_2(C_6H_{15}N_3H)_2)_2$ . For UCR-7GaS-TETA: 10.90 C, 3.06 H, 8.83 N, calculated: 11.50 C, 3.26 H, 8.95 N based on the formula [Ga10S18](C6H18N4H2)2(C6H18N4H3)2/3. For UCR-7GaS-TETA: after heating at 280 °C in Ar, 11.04 C, 1.57 H, 3.25 N; after ion exchange with 2 M NH4Cl, 4.47 C, 2.26 H, 4.44 N. The presence of Ga3+ and Zn2+ ions in T4-containing structures was verified
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